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Characterization of Bulk Hexagonal Boron Nitride Single Crystals Grown by the Metal Flux Technique

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Keywords

Hexagonal boron nitride, hBN, single crystals, photoluminescence, optical absorption

Abstract

The optical and physical properties of hexagonal boron nitride single crystals grown from a molten metal solution are reported. The hBN crystals were grown by precipitation from a nickel-chromium flux with a boron nitride source, by slowly cooling from 1500 °C at 2-4°C/h under a nitrogen flow at atmospheric pressure. The hBN crystals formed on the surface of the flux with an apparent crystal size up to 1 to 2 mm in diameter. Individual grains were as large as 100-200 μm across. Typically, the flakes removed from the metal were 6 to 20 μm thick. Optical absorption measurements suggest a bandgap of 5.8 eV by neglecting the binding energy of excitons in hBN. The highest energy photoluminescence peak was at 5.75 eV at room temperature. The hBN crystals typically had a pit density of $5 \times 10^6 \text{ cm}^{-2}$ after etching in a molten eutectic mixture of potassium hydroxide and sodium hydroxide. The quality of these crystals suggests they are suitable as substrates for two dimensional materials such as graphene and gallium nitride based devices.

Keywords: A1. Characterization; A1. Etching; A2: Growth from solutions; B1. Nitrides

Introduction

Single crystal hexagonal boron nitride (hBN) is the best substrate for supporting graphene, because of its similar crystal structure and lattice constants, ultrasmooth surfaces, low density of surface charge states, and wide energy band gap [1,2]. When it is encapsulated in hBN, graphene's properties are the best, including the highest reported room temperature charge carrier mobility ($>140,000 \text{ cm}^2/\text{V}\cdot\text{s}$) [3] and ballistic transport distances of several microns [2,3].

hBN is also attractive as a substrate for other materials as well. Chan *et al* [4] reported a ten-fold increase in the electron mobility of thin MoS_2 layers on hBN compared to SiO_2 substrates. Similarly, Gehring *et al* [5] saw the surface carrier mobility of the topological insulator BiTe_2Se increase by a factor of three on hBN compared to SiO_2 . hBN has also been proposed as a substrate for atomically thin layers of silicon (silicene) [6] and germanium (germanene) [7]. Since hBN does not have dangling surface bonds, the epitaxial layers it supports are weakly held

and can be mechanically removed, allowing their separation from the substrate, as has been demonstrated for GaN based devices by Makimoto *et al* [8] and Kobayashi *et al* [9].

For all of these applications, single crystals are preferable to polycrystals because the properties of the layer changes with their relative crystallographic orientation to the hBN. That is, grain boundaries can disturb the electrical properties. Ideally, the hBN single crystals should have low defect densities, as these can also adversely affect the layers it supports. Furthermore, high purity hBN crystals are best, to ensure maximum optical transparency over a large range of wavelengths. Thus, this study was undertaken to determine the properties of hBN single crystals that were produced by the flux growth.

There have been relatively few studies on the bulk crystal growth of hBN [10-14]. These employed a variety of solvents for hBN including silicon [10], copper [11], sodium [12], and nickel-chromium [13,14]. Here we adapt the method developed by Kubota *et al* [14] employing a Ni-Cr flux, due to its relatively modest temperature, ability to achieve reasonably fast growth rates, and ability to produce crystals at atmospheric pressure.

Experimental

For these studies, the hBN single crystals were precipitated from a 49wt % nickel-51wt% chromium flux that was used to dissolve the hBN powder source. The mixture was heated to 1500 °C under a flow of nitrogen at atmospheric pressure, and then slowly cooled at 4°C/h to 1250 °C. A more complete description of the crystal growth process was given in our earlier publication [15]. At room temperature, hBN flakes were mechanically peeled off of the solidified metal flux for further study.

Using micro-positioners, a h-BN flake was positioned over a 200-um open aperture in a Au coated surface. Optical images of this structure are shown in Figure 1. Optical transmission data through the flake and 200-um pinhole were measured over a range from 1 to 6 eV using a Cary 5G spectrophotometer in dual-beam mode. The system baseline was calibrated using an identical 200-um aperture without the BN flake.

For PL measurements, a frequency-quadruped Ti-sapphire laser (with photon wavelength of 197 nm, 76 MHz repetition rate, and 100 fs pulse width) was used as an excitation source. The laser beam was focused onto the sample surface with a lens. The PL signal was collected and dispersed by a monochromator (1.3 meters), and then detected by a microchannel-plate photomultiplier tube together with a single photon-counting system [16].

To assess their quality, several hBN crystals were etched in a molten eutectic mixture of sodium hydroxide and potassium hydroxide at 430 °C for 1.5 minutes. Etch pits were assumed to be associated with dislocations in the crystals, as has been shown for other materials in this defect sensitive etchant [17]. Etch pit shapes, sizes, and densities were determined by scanning electron microscopy.

Results

The individual crystals nucleated with their (0001) planes parallel to the metal surface, but with random in-plane orientations. The hBN crystals that formed had an apparent size as indicated by their distinctive boundaries of up to 1 to 2 mm. Closer inspections showed that the individual grain size was up to 100 to 200 μm across. The thickness of the crystals ranged between 6 and 20 microns, as measured by surface profilometry. Frequently, the crystals were cracked, presumably due to contraction as the metal flux solidified, and the differences in the coefficients of thermal expansion of the hBN and the metal flux.

The measured optical transmission spectrum for a h-BN flake is shown in Figure 2. To determine the BN thickness, the oscillations near 2eV were fit to a Fabry-Perot cavity formed by the BN flake. Using 1.85 for the refractive index of BN [18], this provides an estimated thickness of 6.1 μm (see Figure 2b). Knowing the thickness allows us to calculate the spectral absorption coefficient of this h-BN flake, as shown in Figure 3.

The absorption coefficient is related to the bandgap (to first order) by: $\alpha \propto (\hbar\omega - E_g)^n$, where n is $\frac{1}{2}$ and 2 for direct and indirect bandgap materials, respectively. The spectral data are plotted in Figure 4 as a function of a) α^2 and b) $\alpha^{1/2}$ to help identify the band-structure. The quality of linear fit to the data suggest an indirect gap of 5.8 eV. However, the rather wide bandgap of BN causes Coulomb attraction of e-h pairs as well as other effects that may influence the shape of h-BN's bandedge.

The maximum room temperature photoluminescence peak energy for the hBN flake (Figure 5) was at 5.75 eV, which is similar to that reported for the best quality hBN single crystals [19]. Additional peaks were seen at 5.57 eV, which Watanabe *et al* [20] attribute to stacking disorder, and at 4.45 eV, which is possibly associated with impurities.

Etching produced two types of hexagonal pits: those with pyramidal sidewalls converging to a point at the bottom of the pit, and those with a flat surface at the bottom of the pit (Figure 6 a). The flat bottom pits may indicate that the dislocation came to an end on the flat plane, or its path moved horizontal, in the plane. The etch pit density was typically $5 \times 10^6 \text{ cm}^{-2}$ in the broad areas of the individual grains. Not unexpectedly, the etch pit densities were higher at grain boundaries.

Discussion

The measurements made in this study demonstrate that the hBN produced by the flux method are of excellent structural and optical quality. The method is relative simple, requiring neither an excessive temperature nor pressure, suggesting the technique can be scaled to produce large crystals. It can produce hBN crystals that are relatively thick in comparison hBN prepared by thin film techniques.

Whether hBN has a direct or indirect bandgap has been debated in the literature. From *ab initio* calculations, Blase *et al* [21] predicted that bulk hBN has an indirect band gap of 5.4 eV.

On the basis of ultraviolet luminescence at 5.765 eV at room temperature and intrinsic fundamental absorption spectra with peaks between 5.822 eV to 5.968 eV at low temperature (8 K) with an s-like exciton structure, Watanabe *et al* [19] claimed its bandgap is direct. Based on the intense photoluminescence at low temperature from epitaxial hBN layers deposited on sapphire substrates, Majety *et al* [16] also supported the finding that hBN is a direct bandgap semiconductor. Using state-of-the-art all-electron GW approximation, Arnaud *et al* [22] asserted hBN is in fact an indirect band gap semiconductor. They estimated the minimum indirect and direct gaps as 5.95 eV and 6.47 eV respectively. The absorption measurement results shown in Fig. 4 seem to suggest that hBN has an indirect bandgap.

Liu *et al* [23] and Gao [24] both predict that variations in the stacking sequence of layers modify the band gap magnitude and whether it is direct or indirect. Specifically, Liu *et al* [23] writes there exists a metastable structure of hBN, which corresponds to a glide distance of B-N bond length (1.446 Å) of all B layers of an ABAB . . . stacking aligning in a direction of in-plane B-N bond that is indirect. This configuration is quite stable and its total energy is only slightly above the ground state and forms a direct band with an energy gap of about 0.6 eV smaller than that of the ground state. Perhaps the structure of layer stacking of hBN may be different for materials grown by different methods, which could contribute to the debate regarding whether hBN has a direct or indirect bandgap. Our experimental results support a gap of 5.8 eV. Measured absorption at energies below the gap up to 5.5 eV show an exponential fit with energy suggesting Urbach tail states due to the disorder (polycrystallinity, etc.). The transition region between Urbach tail states and the indirect band edge could include states from dangling bonds, similar to those found in micro- and poly-crystalline Si films. However, despite the debating issue concerning the bandgap, clearly, the band-edge emission efficiency in hBN is unusually high due to its layered structure [25]. The layer structured hBN provides a natural 2D system which results in a sharp “step”-like density of states and increase in the exciton binding energy and oscillator strength.

For the bulk hBN crystals, the grain size is larger and their defect densities are much lower than those typically found in the deposited thin films. For example, for their thin films, Tay *et al* [26] reported a typical hBN domain size that was less than 1.0 square microns. Using transmission electron microscopy, Levendorf *et al* [27] established their hBN grain size was on the order of 1 to 10 microns. Lee *et al* [28] observed the domain size of their hBN layers were determined by the grain size of the nickel foil they employed, on the order of 100 microns or less. For the hBN samples grown in this study, the grain size regularly exceeded 100 microns, and the potential exists for increasing the size further by modifying and improving the process.

Conclusions

Hexagonal boron nitride single crystals of good quality were produced from a nickel-chromium molten metal flux at atmospheric pressure. Optical absorption, photoluminescence, and defect selective etching demonstrates hBN's wide energy bandgap (5.8 eV), high energy peak (5.76 eV), and relatively low etch pit densities respectively. Additional studies are needed to eliminate cracking, increasing the crystal size, identifying the impurities present, and to further

reduce the defect densities. With these improvements, this process may produce large area hBN crystals.

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Figure Captions

Figure 1. Confocal microscope images of the h-BN flake mounted on a 200-um aperture in a Au coated surface using a) white light, and b) 400-nm laser light.

Figure 2 – a) Optical transmission spectrum of a h-BN flake from 1-6eV, along with b) a fit to the Fabry-Perot oscillations providing an 6.1-um estimated thickness.

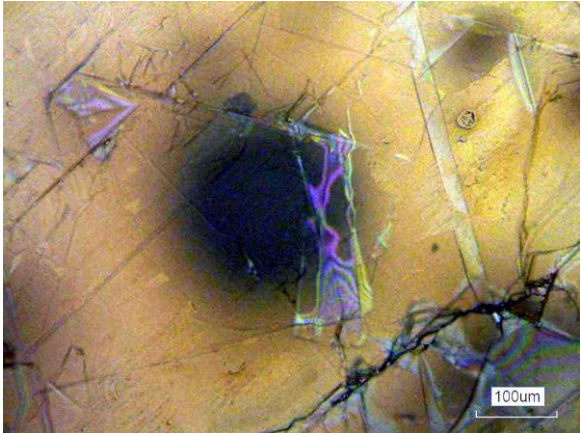
Figure 3 – Absorption coefficient for 6.1-um thick h-BN flake.

Figure 4 – Linear fit to spectral data assuming proportionality to α^2 and $\alpha^{1/2}$ to estimate the type of bandgap for the h-BN flake.

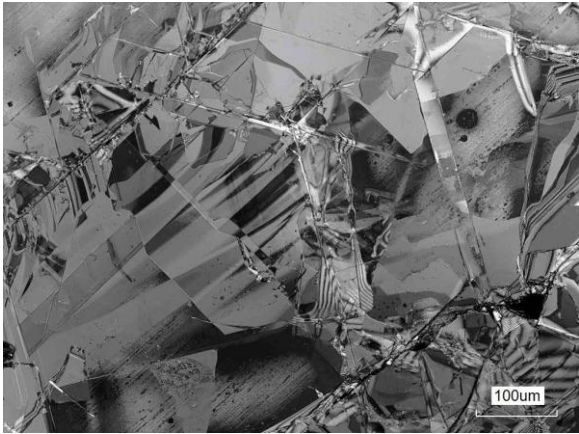
Figure 5 – Room temperature photoluminescence from a hBN crystal.

Figure 6 Scanning electron microscope image of an individual pit formed by etching an hBN crystal in a eutectic mixture of potassium hydroxide and sodium hydroxide. The hexagonal pits had either pointed or flat bottoms (Scale bar = 4.0 microns).

Figure 1

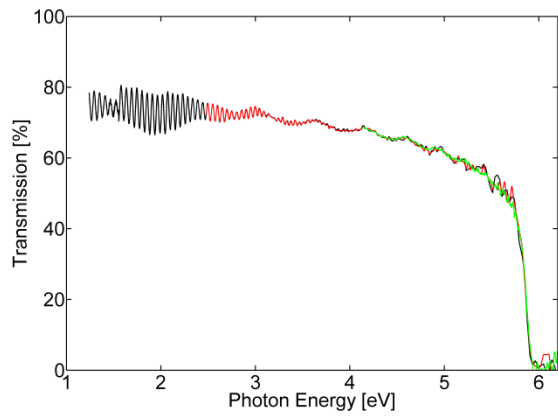


(a)

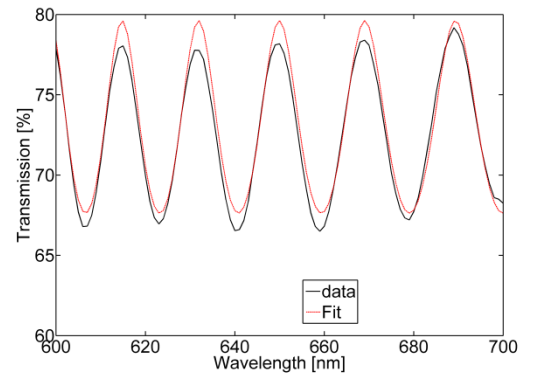


(b)

Figure 2.



(a)



(b)

Figure 3.

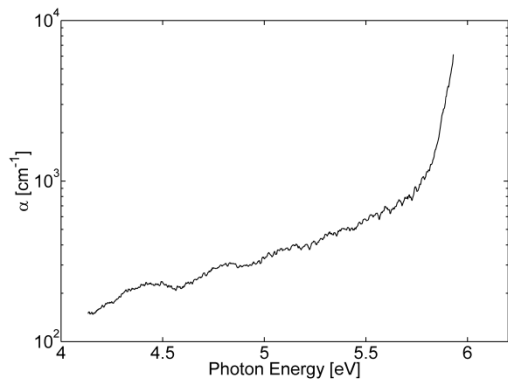


Figure 4

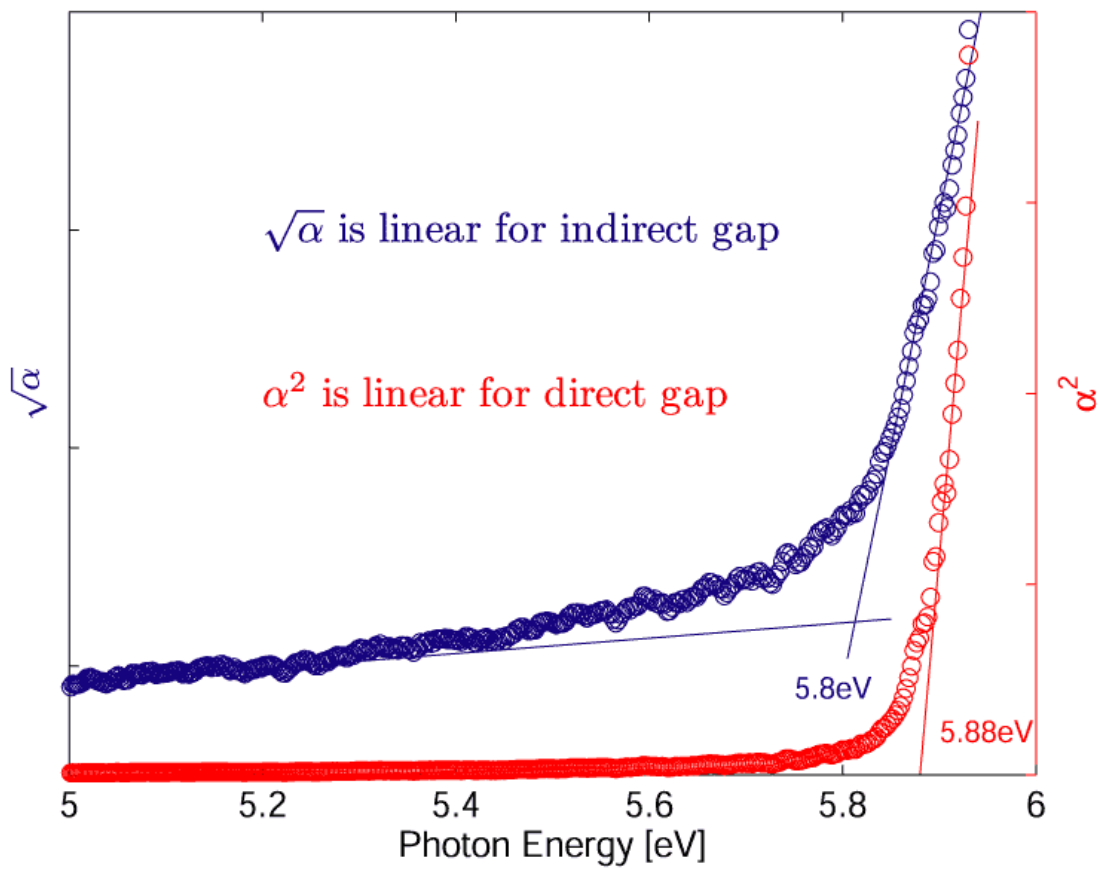


Figure 5

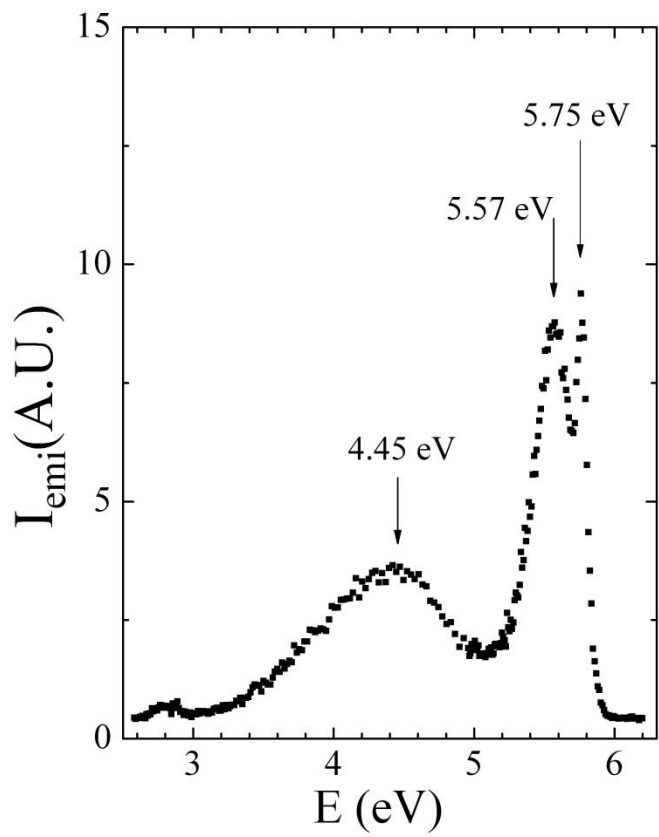


Figure 6

